Some Thermodynamic, Structural and Behavioral Aspects of Materials Accentuating Non-crystalline States

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22. DIFFUSION STRUCTURAL DIAGNOSTICS OF SELF-IRRADIATED AMORPHOUS MINERALS

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22. 1. Diffusion structural diagnostics of solids

Diffusion structural diagnostics of solids consists in the characterization of diffusion properties of solid samples by using inert gases as traces. Changes in surface morphology and microstructure of the solid samples during thermal treatments and changes due to chemical, mechanical or radiation interactions can be studied by the diffusion structural diagnostics methods.
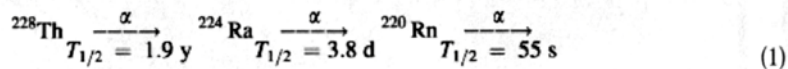
Emanation thermal analysis (ETA) [1-3], based on the measurement of the radon release from samples, is one of the methods used in the diffusion structure diagnostics of solids.

As most of the solids to be investigated do not naturally contain atoms of radon it is necessary to introduce the radon atoms in the samples prior to the ETA measurements.

To introduce the radioactive trace into the solids, the samples are labeled by parent radio-nuclides ^{228}Th and ^{224}Ra, serving as a quasi-permanent source of radon atoms ^{220}Rn. The used specific activity of the parent radionuclide ^{228}Th is in the order 10^5 Bq per gram of the sample.

22.1.1. Use of recoil energy of α-decay for radon atoms implantation into solids

Atoms of ^{220}Rn radon radionuclide are formed by a spontaneous α-decay of ^{228}Th and ^{224}Ra radio-nuclides according to scheme Eq.1

$$^{228}\text{Th} \xrightarrow[T_{1/2}\,=\,1.9\,\text{y}]{\alpha} {}^{224}\text{Ra} \xrightarrow[T_{1/2}\,=\,3.8\,\text{d}]{\alpha} {}^{220}\text{Rn} \xrightarrow[T_{1/2}\,=\,55\,\text{s}]{\alpha} \tag{1}$$

and the ^{220}Rn atoms can be introduced into the solid owing to the recoil energy of the spontaneous α-decay (85 keV per atom). The samples can be labelled by using an adsorption of traces of ^{228}Th as nitrate from a solution.

Due to the energy of the spontaneous α-decay of ^{228}Th and ^{224}Ra radio-nuclides the atoms of ^{220}Rn can penetrate into the sample several tens of nanometers from the surface depending on the composition of the materials .The values of the maximum penetration depths of ^{220}Rn were determined by the Monte Carlo method using TRIM code [4], e.g. for SiO_2 :65.4 nm, for zircon (Zr SiO_4) : 60 nm and brannerite mineral ($U_{1-x}Ti_{2+x}O_6$): 60 nm.

22.1.2. Trapping and release of radon atoms from solids

Radon atoms can be trapped in solids at structure defects such as vacancy clusters, grain boundaries and pores. The structure defects in the solids can serve both as traps and as diffusion paths for radon atoms.

The radon formed by spontaneous α-decay of ^{224}Ra may escape from the solid either by recoil energy ejection or by diffusion. The term emanation rate, E, has been

used to express the release of radon from solids. It is defined as the ratio of the radon release rate to the rate of radon formation by the spontaneous α-decay of ^{228}Th and ^{224}Ra in the investigated solids.

It has been determined experimentally (in relative units) as $E = A_\alpha/A_{\text{total}}$, where A_α is the α-radioactivity of radon released in unit time from the labeled sample and A_{total} is the total γ-radioactivity of the labeled sample. The A_{total} value is proportional to the rate of radon formation in the sample.

In the evaluation of the radon release from solids several mechanisms have been supposed, namely the radon release by recoil mechanism, the diffusion in open pores, and the volume diffusion mechanism.

The experimentally obtained values of the emanation rate, E , can be considered as:

$$E = E(\text{recoil}) + E(\text{pores}) + E(\text{solid}). \tag{2}$$

The emanation rate due to recoil, E (recoil), can be expressed as

$$E(\text{recoil}) = K_1 S_1 \tag{3}$$

where K_1 is a temperature independent constant, proportional to the penetration depth of recoiled radon atoms in solids investigated and S_1 is external surface area of sample particles. The path of recoiled atoms of radon is dependent on the "nuclear stopping power" of the sample material.

The emanation rate due to diffusion in pores, E (pores), is expressed as

$$E(\text{pores}) = K_2 S_2 \tag{4}$$

where K_2 is a constant that depends on temperature and S_2 is internal surface area of the sample depending on the surface of open pores, cracks and intergranular space . The emanation rate due to volume diffusion mechanism, E (solid), is expressed as

$$E(\text{solid}) = K_3 \exp(-Q/2\mathbf{R}T) S_3 \tag{5}$$

where K_3 is a constant related to the atomic properties of the lattice, Q is the activation energy of Rn diffusion in the solid, S_3 is surface area, $\mathbf{R}$ is molar gas constant, and T is temperature.

The growth of the emanation rate values, E (T), may characterize an increase of the surface area of interfaces, whereas a decrease in the E(T) may reflect processes like closing up structure irregularities that serve as paths for the radon migration, closing pores and/or a decrease in the surface area of the interfaces [3, 5].

22.1.3. Measurement of radon release from solids

During the ETA measurements, a constant flow of the carrier gas (air, nitrogen, or another gas) has been used to take the radionuclide of radon of ^{220}Rn released by the sample into the detector of α-activity of radon (semiconductor detectors) [6].Figure 1 depicts a scheme of the apparatus for radon measurement used in the emanation thermal analysis.

In the study of the self- irradiated amorphous materials the ETA measurements were carried out by using modified NETZSCH DTA-ETA equipment, Type 404. Details of the measurements and the data treatments are described elsewhere [2,3,6]. During the ETA measurements the samples were heated at the rate of 6 K.min^{-1} in air

(zircon sample) and argon (brannerite sample). The specific activity of the labelled samples was 10^5 Bq.g^{-1}. The used amounts of the samples were 0.02 and 0.1 g respectively.
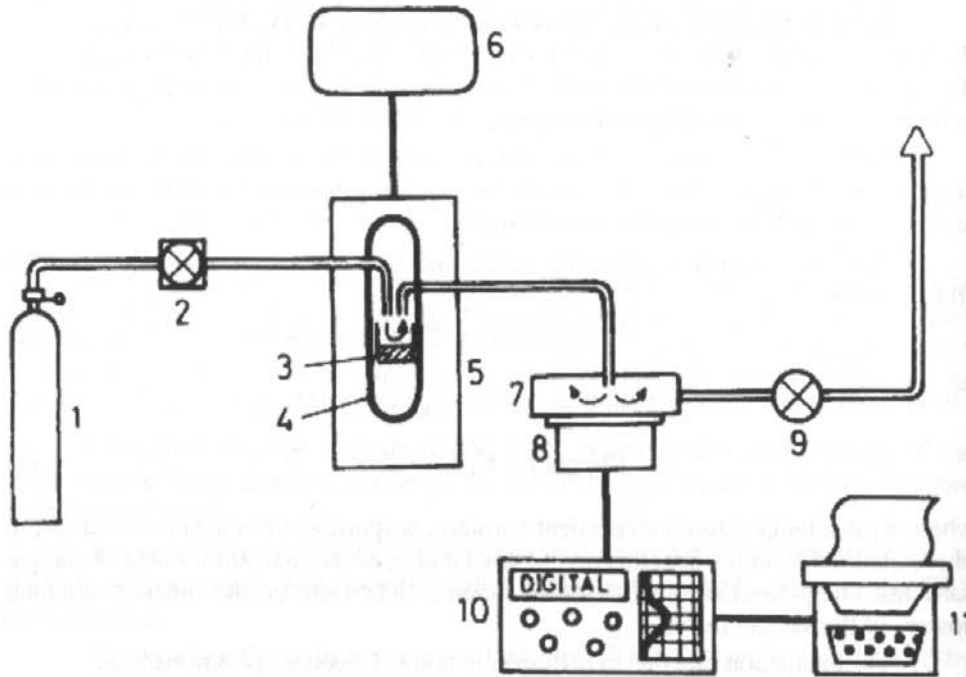

Fig. 1 *Scheme of the apparatus for emanation thermal analysis: 1—gas supply; 2—gas flow stabilizer; 3—labeled sample; 4—sample holder; 5—furnace; 6—temperature controller; 7—measuring chamber; 8—radioactivity detector; 9—flow- rate meter; 10—counts-meters; 11—data processor and printer (plotter). Below the photo of the updated Netzsch DTA-ETA 404 (digitalized equipment commercially produced by Netzsch Gerätebau GmbH, Selb, Germany).*
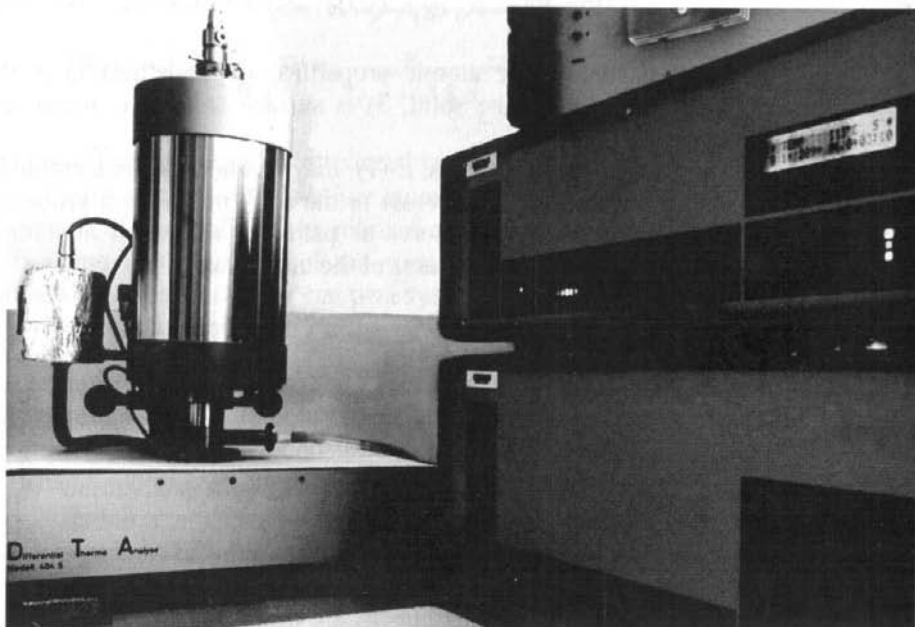
22. 2. The ETA application in the diffusion structural diagnostics of materials

Equipment for the emanation thermal analysis (ETA) was developed in the 1960s at the *Nuclear Research Institute, Řež.* (Czech Republic) [1,2]. Since that time the ETA method was used in various investigations, e.g. the re-crystallization of solids, annealing of structure defects and changes in the defect state of both crystalline and amorphous solids, sintering, phase changes, the characterization of surface and morphology changes accompanying chemical reactions in solids and on their surfaces, including the thermal degradation, solid–gas, solid–liquid, and solid–solid interactions [7-11]. The ETA made it possible to reveal even fine changes in poorly crystalline or amorphous solids. Differences in the morphology and behavior of samples prepared by the sol–gel technique under different conditions were revealed by the ETA.

Changes in defects annealing and pore sintering of the samples were characterized by using the ETA results under in situ conditions of their heat treatments. The determination of optimized conditions for the preparation and thermal treatments of advanced ceramic materials was achieved [10,11]. By this way the ETA results contributed to the solution of practical tasks in the materials technology.

Recently, this method made it possible to characterize the thermal stability of ceramic materials designed for the immobilization/encapsulation of high level radioactive waste [12]. Moreover, the thermal stability of self- irradiated amorphous minerals that serve as natural analogues of the ceramic matrices was evaluated by using the emanation thermal analysis.

22.2.1. Thermal behavior of amorphous zircon mineral

Natural zircon mineral (general formula $ZrSiO_4$), containing an average concentration up to 0.4 % of uranium and 0.2 % of thorium, has attracted much interest from both fundamental and technological view points. The α-particles and heavy recoil nuclei released during the decay of radioactive impurities (typically ^{238}U, ^{235}U and ^{232}Th) interact with the surrounding crystalline matrix displacing atoms from their equilibrium positions [13]. Over geological periods of time this process disrupts the crystalline order to such a point that specimens covering all the stages from fully crystalline to amorphous can be found, depending on the uranium/thorium content.

Understanding the radiation effects in crystalline zircon and the determination of the structure of the aperiodic state are essential to ensure the reliability of zircon based ceramics for nuclear waste disposition [13,14]. During nuclear disintegration, the emission of the α-particle is accompanied by a recoil nucleus. Amorphization taking place in natural zircon is called metamictization. The α-particles have an energy of 4-6 MeV, and almost all the energy is dissipated by the ionization processes. It is believed that various isolated defects, such as Frenkel pairs, are formed along their paths. A number of studies have been devoted to structural changes of zircon under irradiation, in particular to understanding the amorphization and/or metamictization process [15]. This process can lead to an increased solubility and fracturing [16].

Ceramic forms used in the encapsulation of nuclear waste are subjected to a similar transformation, with the corresponding variation of their physical and chemical properties. An understanding of radiation effects in crystalline zircon and a determination of the structure of the aperiodic state are essential to ensure the reliability of zircon and related ceramics for nuclear waste disposition [13, 14].

Zircon ceramics can incorporate significant amount of UO_2, PuO_2 or ThO_2 in a

solid solution with ZrO_2. The zircon undergoes an amorphization promoted by α-decay events of radiogenic elements. During the nuclear disintegration, the emission of an α-particle is accompanied by a recoil nucleus and ballistic collisions of the recoil nucleus cause displacement cascades. A number of studies have been devoted to the evolution of amorphous zircon under irradiation, in particular to understanding of the metamictization process [15].

Natural zircon mineral sample characterized by ETA was from the locality in New Mexico, USA. The sample was X-ray amorphous [16].

Figure 2 below shows ETA results of the zircon mineral sample measured on heating (curve *1a*) in air flow in the temperature range 20 – 1100 °C and subsequent cooling (curve *1b*). The increase of the emanation rate, *E*, observed in the temperature range of 170 – 250 °C characterized the diffusion mobility of radon atoms along surface cracks and other subsurface defects, the subsequent decrease of the *E* values in the range 250 – 420 °C can be ascribed to healing the surface and subsurface defects.
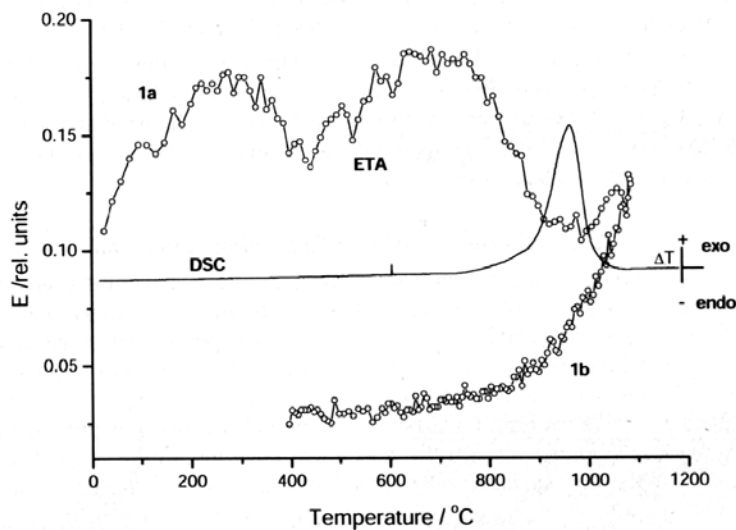

Fig. 2 *ETA results (points) of natural self irradiated zircon mineral sample measured on heating (curve 1a) and subsequent cooling (curve 1b) in air in the range 20-1100 °C. DSC results measured on the sample heating are depicted as the full line curve*

We supposed that the increase of the emanation rate, *E*, in the range 420 – 750 °C is due to the radon diffusion along structure irregularities in the amorphous zircon. The phase transformation of initially amorphous zircon was characterized by the decrease of the emanation rate values *E* in the range 750 – 950 C. From ETA results of the amorphous zircon mineral sample measured on heating to 1200 °C and subsequent cooling to room temperature it followed that the microstructure changes taking place in the sample on heating were irreversible.

The results of DSC measured on a parallel sample of amorphous/metamict zircon are demonstrated in Figure 2 as the full line curve. The transformation of amorphous zircon to the crystalline zircon was characterized by a DSC exothermal effect with the maximum at 918 °C.

22.2.2. Thermal behavior of amorphous brannerite mineral

Brannerite mineral (general formula $U_{1-x}Ti_{2+x}O_6$) has been found in nature as amorphous due to α-decay damage caused by high content of U, Th. The formula of natural brannerite can also be written $(U, Th)_{1-x}Ti_{2+x}O_6$. The natural brannerite generally contains impurity elements like Pb, Ca, Th, Y and rare earth elements (REE) on the U-site and Si, Al and Fe on the Ti-site. The brannerite is a minor phase in titanate-based ceramics designed for the geological immobilization of surplus Pu [17,18]. Therefore, it was of interest to investigate the thermal behaviour of brannerite mineral as a natural analogue of the brannerite ceramics to be used for immobilization of hazardous radioactive elements.

The diffusion structural diagnostics based on the results of emanation thermal analysis (ETA) made it possible to characterize the annealing of the structure irregularities in the brannerite mineral sample on heating to various temperatures up to 1200°C. Natural brannerite mineral was from the locality El Cabril mine near Cordoba, Spain. The sample was X-ray amorphous and contained Ca, Pb and other impurity elements [17].

Figure 3 shows the ETA results of the mineral brannerite sample measured during heating in argon in the range 20 – 1200 °C and subsequent cooling. The increase of emanation rate, E, observed on the sample heating in the range of 40 – 300 °C characterized the diffusion mobility of radon atoms along surface cracks and other subsurface defects to depth of 60 nm.

The slight decrease of $E(T)$ observed in the temperature range of 400 – 500 °C (curve *1a*, Figure 3) was ascribed to healing surface cracks and voids. The decrease of the emanation rate $E(T)$ observed on the ETA curve in the range 800 – 880 °C, corresponding to the healing microstructure irregularities, was considered as a first step of the formation of crystalline brannerite. The increase of $E(T)$ observed in range 900 – 965 °C followed by the sharp decrease of $E(T)$ in the range 970 – 1020 °C indicated the formation of the crystalline brannerite phase, as confirmed by XRD spectroscopy [16]. From ETA results of the brannerite mineral sample measured on heating up to 1200 °C and subsequent cooling it followed that the microstructure changes on sample heating are irreversible.
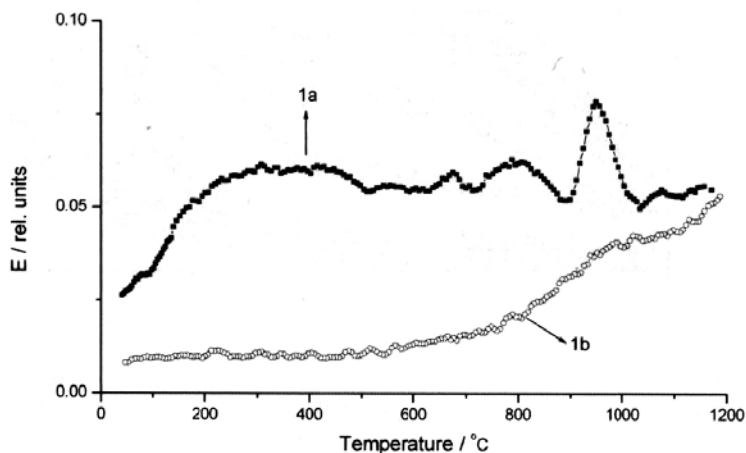

Fig. 3. ETA results of self-irradiated amorphous brannerite mineral sample measured on heating (curve 1a) and subsequent cooling (1b) in argon in the range 2 –1200 °C.

The release of CO_2 was detected by mass spectrometry of evolved gases in the temperature range 700-800°C [16] due to the thermal degradation of minor carbonate containing components of the sample. The release of CO_2 gave rise to the sample porosity [16].

Microstructure differences of the self-irradiated brannerite mineral samples before and after heating to 1200°C are characterized by the SEM micrographs (Figure 4).The crystallization of initially amorphous brannerite was characterized by the ETA as changes of radon the diffusion rate in the range 900-1020 °C (Figure 3), in agreement with X-ray diffraction results [17-19].
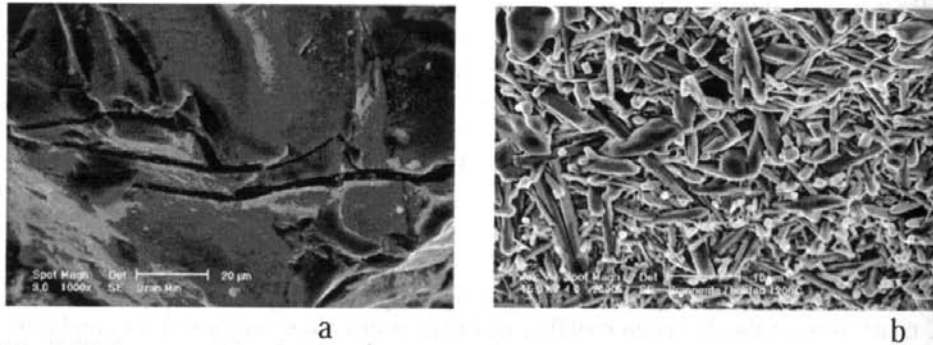

a   b

Fig. 4 *SEM micrographs of amorphous brannerite mineral sample: (a) as received (b) heated to 1200 °C*

It was of interest to investigate the self-irradiated brannerite mineral during "step by step" heating and subsequent cooling of the sample to the temperatures of 300, 550, 750, 880, 1020 and 1150 °C, respectively. Results of ETA measured by the "step by step" heating runs (Figure 5) made it possible to compare the annealing of microstructure irregularities of the sample in the selected temperature intervals .
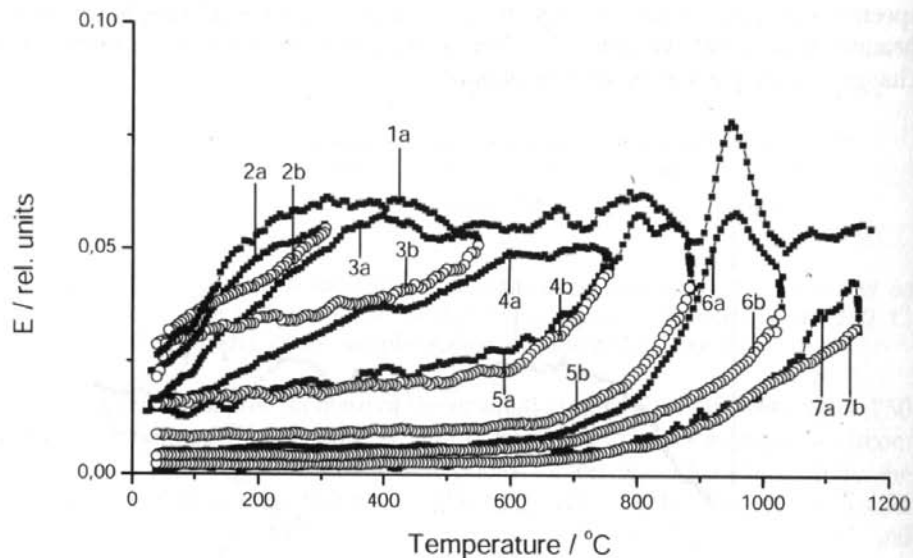

Fig. 5 *ETA results of the self-irradiated amorphous brannerite mineral sample measured on heating and subsequent cooling in argon in the range 20-1150 °C: curve 1a corresponds to the "as received" sample measured during heating from 20 to 1150 °C, curves 2a/2b, 3a/3b, 4a/4b, 5a/5b, 6a/6b and 7a/7b were measured with a parallel samples pre-heated to the temperatures of 300, 550, 750, 880 and 1020 °C, respectively.*

As it follows from the ETA results in Figure 5, the "step by step" heating of the sample to these temperatures caused a decrease of the amount of structure irregularities serving as radon diffusion paths.

A good reproducibility of the ETA results measured on heating from 20 to 300 °C is obvious from the comparison of the results in Figure 5, curve *2a* and curve *1a*. The ETA curves *3a/3b*, *4a/4b* and *5a/5b* characterized the thermal behaviour of brannerite sample pre-heated to 300 and 750 °C, respectively. The increase of the emanation rate, E, in the temperature range of 20 – 360 °C, due to the diffusion of radon along micropores in the sample, was followed by the decrease of E, characterizing the partial healing of voids and structure irregularities that served as diffusion pathways for radon.

The ETA curves *6a/6b* in Figure 5 characterized the thermal behaviour of the sample pre-heated to 880 °C. As already observed by curve *1a* the amount of structure irregularities serving as radon diffusion paths further diminished in the sample pre-heated to 880 °C. The decrease of the emanation rate on sample observed on heating in the range of 970 – 1020 °C indicated the next step of the formation of crystalline brannerite. A good reproducibility of the ETA measurements can be seen from the temperature coincidence of the effects on the curve *1a* and curve *6a* in Figure 5.

From curves *7a/7b* characterizing the thermal behaviour of the sample pre-heated to 1020 °C, it is obvious that after the pre-heating the sample to this temperature an irreversible crystallization of amorphous self-irradiated brannerite mineral took place.

From Figure 5 it is obvious the amount of structure irregularities serving as radon diffusion paths further diminished and the radon permeability in the pre-heated brannerite samples decreased with the temperature used for pre-heating of the samples.

Values of the emanation rate, E_{RT}, measured at room temperature before and after each heating run were used for the assessment of the relative changes of the surface area affected by the heat treatments used. The E_{RT} values summarized in Table 1 are in agreement with our considerations of the annealing of surface area and subsurface irregularities.

Table 1 Microstructure defects characteristics of self-irradiated amorphous brannerite mineral sample pre-heated to various temperatures

ETA curves measured on heating/cooling	Temperature of sample pre-heating	Defect amount characteristics ξ^*	E_{RT} [rel. units]	$\Delta\xi^{**}$ [%]
Curves 1a/1b ,Fig.3	as received	38.1	0.026	100
Curves 2a/2b, Fig.5	as received	0.41	0.023	1.08
Curves 3a/3b, Fig.5	300 °C	3.82	0.017	10.02
Curves 4a/4b, Fig.5	550 °C	10.26	0.015	26.93
Curves 5a/5b, Fig.5	750 °C	13.62	0.014	35.75
Curves 6a/6b, Fig.5	880 °C	6.30	0.005	16.54
Curves 7a/7b ,Fig.5	1020 °C	0.98	0.001	2.57

$$* \; \xi(T_{max}) = \int_{T_{min}}^{T_{max}} E(T)_{heating}\, dT - \int_{T_{min}}^{T_{max}} E(T)_{cooling}\, dT \qquad ** \; \Delta\xi = \frac{\xi_n}{\xi_1} \times 100 \quad [\%]$$

From the temperature dependences of the emanation rate, $E(T)$, measured during heating to selected temperatures and subsequent cooling, the decrease in the amount of radon diffusion paths was assessed. To this aim we used the parameter ξ defined in Eq. 6 as:

$$\xi(T_{max}) = \int_{T_{min}}^{T_{max}} E(T)_{heating}\, dT - \int_{T_{min}}^{T_{max}} E(T)_{cooling}\, dT \qquad (6)$$

Moreover, values of $\Delta\xi$ (see Eq. 7) were calculated with the aim to compare the amounts of the annealed microstructure defects during the "step by step" heating of the sample. The difference of integrals used for the assessment of the amount of the microstructure defects can be expressed as $\Delta\xi$ defined as

$$\Delta\xi = \frac{\xi_n}{\xi_1} \times 100\,[\%] \qquad (7)$$

As it followed from values of ξ and $\Delta\xi$ summarized in Table 1, the most significant decrease of the structure irregularities serving as diffusion paths for radon diffusion was annealed prior to the crystallization of the sample in the range of 970 – 1020 °C.

Fig. 6 depicts a comparison of the relative amount of structure irregularities, expressed by parameter ξ, that were annealed during heat treatments to the selected temperatures.

It was shown that the emanation thermal analysis revealed differences in the amount of structure irregularities that served as radon diffusion paths in the amorphous brannerite samples. Additional information about thermal behaviour of self-irradiated amorphous minerals was obtained by using the diffusion structural diagnostics.
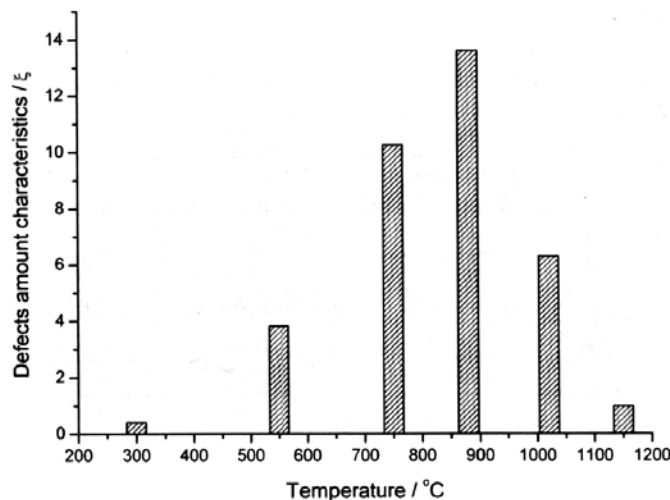

Fig. 6. *Relative amounts of structure irregularities in the self-irradiated amorphous brannerite sample healed in the heating runs to temperatures 20 - 300, 20 - 550, 20 - 750, 20 - 880, 20 - 1020 and 20 - 1150 °C. Parameter ξ was used to characterize the amount of structure irregularities of the following samples preheated to 300, 550, 750, 880 and 1020 °C respectively.*

References 22

1. V. Balek, *Emanation thermal analysis.* Thermochim. Acta , 22 (1978) 156.

2. V. Balek, *Emanation thermal analysis and its application potential.* Thermochim Acta, 192 (1991) 1.

3. V. Balek, J. Šubrt, T. Mitsuhashi, I.N. Beckman and K. Györyová, *Emanation thermal analysis: ready to fulfill the future needs of materials characterization.* J. Therm. Anal. Cal. 67 (2002) 15

4. J.F. Ziegler, J.P Biersack, and U. Littmark, *The Stopping and Range of Ions in Solids,* Pergamon Press, New York, 1985.

5. I.N. Beckman and V.Balek, *Theory of emanation thermal analysis XI. Radon diffusion as the probe of microstructure changes in solids.* J. Therm. Anal. Cal. 67 (2002) 49.

6. W.D. Emmerich, V.Balek , *Simultaneous application of DTA, TG, DTG, and emanation thermal analysis.* High.Temp.-High Press.5 (1973) 67

7. V. Balek, J. Tölgyessy: *Emanation thermal analysis and other radiometric emanation methods,* in Wilson and Wilson's Comprehensive Analytical Chemistry, Part XIIC, G. Svehla Ed.; Elsevier Science Publishers, Amsterdam, 1984, 304pp.

8. V. Balek, M.E. Brown, *Less common techniques,* in: Handbook on Thermal analysis and calorimetry, Vol.1 Chapter 9, Edited by M.E. Brown, ELSEVIER SCIENCE B.V., 1998 p.445.

9. V. Balek , *Characterization of high-tech materials by means of emanation thermal analysis.* J. Therm. Anal. 35(1989) 405

10. V. Balek , J. Šesták , *Use of emanation thermal analysis in characterization of superconducting $YBa_2Cu_3O_x$.* Thermochim. Acta ,133(1988) 23

11. V. Balek, J.L. Pérez-Rodriguez, L.A. Pérez-Maqueda, J. Šubrt, J. Poyato, *Thermal behaviour of ground vermiculite.* J.Therm. Anal. Cal., 88 (2007) 819.

12. V. Balek , Y. Zhang V. Zeleňák, J. Šubrt, I.N. Beckman, *Emanation thermal analysis study of brannerite ceramics for immobilization of hazardous waste.* J. Therm. Anal. Cal., 92 (2008) 155

13. S. Ríos, T. Boffa-Ballaran, *Microstructure of radiation-damage zircon under pressure.* J. Appl. Cryst., 36 (2003) 1006

14. R. Devanathan, L. R. Corrales, W.J. Weber, *Molecular dynamics simulation of disordered zircon.* The American Physical Society, Phys. Rev. B, 69 (2004) 064115-1

15. P. Carrez, Ch. Forterre, D. Braga, H. Leroux, *Phase separation in metamict zircon under electron irradiation.* Nucl. Inst. Meth. in Phys.Res., B 211 (2003) 549

16. L.M. Wang, R.C. Ewing, *Detailed in situ study of ion beam-induced amorphization of zircon.* Nucl. Inst. Methods in Phys. Res., B 65 (1992) 324

17. V. Balek, E.R.Vance, V. Zeleňák, Z. Málek, J. Šubrt, *Use of emanation thermal analysis to characterize thermal reactivity of brannerite mineral.* J. Therm. Anal. Cal. ,88 (2007) 93

18. J. Lian, L.M. Wang, G. R. Lumpkin, R.C. Ewing, *Heavy ion irradiation effects of brannerite-type ceramics.* Nucl. Inst. Methods in Phys. Res. B 191(2002)565

19. Y. Zhang, G.R. Lumpkin, H. Li, M.G. Blackford, M. Colella, M.L. Carter, E.R. Vance, *Recrystallisation of amorphous natural brannerite through annealing: The effect of radiation damage on the chemical durability of brannerite.* J. Nucl. Mat., 350, (2006) 293.